

High-Temperature Ferrimagnetism Driven by Lattice Distortion in Double Perovskite $\text{Ca}_2\text{FeOsO}_6$

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Supporting Information

ABSTRACT: 5d and 3d hybrid solid-state oxide $\text{Ca}_2\text{FeOsO}_6$ crystallizes into an ordered double-perovskite structure with a space group of $P2_1/n$ with high-pressures and temperatures. $\text{Ca}_2\text{FeOsO}_6$ presents a long-range ferrimagnetic transition at a temperature of ~ 320 K (T_c) and is not a band insulator, but is electrically insulating like the recently discovered $\text{Sr}_2\text{CrOsO}_6$ ($T_c \sim 725$ K). The electronic state of $\text{Ca}_2\text{FeOsO}_6$ is adjacent to a half-metallic state as well as that of $\text{Sr}_2\text{CrOsO}_6$. In addition, the high- T_c ferrimagnetism was driven by lattice distortion, which was observed for the first time among double-perovskite oxides and represents complex interplays between spins and orbitals. Unlike conventional ferrite and garnet, the interplays likely play a pivotal role of the ferrimagnetism. A new class of 5d–3d hybrid ferrimagnetic insulators with high- T_c is established to develop practically and scientifically useful spintronic materials.

Solid-state perovskite oxides, with the general structure ABO_3 , with unpaired d-electrons show scientifically and practically useful electromagnetic properties such as ferromagnetism, multiferroicity, and colossal magnetoresistance, depending on the combinations of A (alkali metal, alkali earth metal, rare earth metal, or solution of those) and B (transition metal). Most of the useful properties appear when B is a 3d element, probably because unpaired 3d electrons strongly correlate with each other or with oxygen 2p electrons in the perovskite lattice.¹ For example, the on-site Coulomb interaction U , which roughly outlines the degree of electronic correlations is in the range of 2–10 eV for 3d perovskite oxides,^{2,3} while U of 5d perovskite oxides is usually < 1 eV.^{2,3}

Recently, 5d(4d)–3d hybrid solid-state oxides have attracted great attention due to the half-metallic (HM) nature found in double perovskite oxides $\text{Sr}_2\text{FeMoO}_6$ ^{4–6} and $\text{Sr}_2\text{FeReO}_6$,^{7–9} which exhibit spin-polarized transitions at temperatures greater than 400 K. The double perovskite oxide is usually denoted as $\text{A}_2\text{BB}'\text{O}_6$, where B and B' atoms are ordered, doubling the unit cell of ABO_3 . In the HM state, 3d– t_{2g} and 5d(4d)– t_{2g} electrons

in the B and B' sites, respectively, play a pivotal role in the highly spin-polarized conduction, which was argued to result from a generalized double exchange mechanism.^{10,11} The high temperatures imply possible developments of novel spintronic applications, which may work without cooling. Furthermore, additional double perovskite oxides have been suggested to be HM at room temperature.^{7,12–23} Therefore, prospects for spintronics applications are very high and enormous efforts have been made not only to develop HM properties, but also to explore novel 5d(4d)–3d hybrid properties.

The double perovskite oxide $\text{Sr}_2\text{CrOsO}_6$, which was synthesized in 2007, showed a ferrimagnetic (FIM) transition at T_c of ~ 725 K,²⁴ the highest reported T_c of the perovskite and double perovskite oxides.¹⁶ However, the generalized double exchange mechanism was unable to account for the 725 K transition because $\text{Sr}_2\text{CrOsO}_6$ is a multiorbital Mott insulator and the FIM state is thus distinguishable from the HM state.²⁵ The high- T_c FIM transition was novel; therefore, an additional 5d–3d hybrid high- T_c FIM oxide was highly desired to elucidate the mechanism of the high- T_c transition.

Here we report a novel double perovskite oxide, $\text{Ca}_2\text{FeOsO}_6$, synthesized using a high-temperature and pressure. It crystallizes into a monoclinic double perovskite structure like many double perovskite oxides. Notably, it shows an FIM transition at a T_c of 320 K. It is not a band insulator, but is highly insulating electrically. Thus, the explanation for the high- T_c FIM transition of $\text{Ca}_2\text{FeOsO}_6$ is likely similar to that of $\text{Sr}_2\text{CrOsO}_6$.

Polycrystalline $\text{Ca}_2\text{FeOsO}_6$ was synthesized by solid-state reaction under high-pressure, and the final product was characterized using synchrotron X-ray diffraction (SXRD; see Supporting Information for details). The SXRD study determined the crystal structure of $\text{Ca}_2\text{FeOsO}_6$ to be monoclinic with a space group of $P2_1/n$, as was reported for many B-site ordered double perovskites.^{26,27} The refined SXRD patterns are shown in Figure 1. Generally in a double

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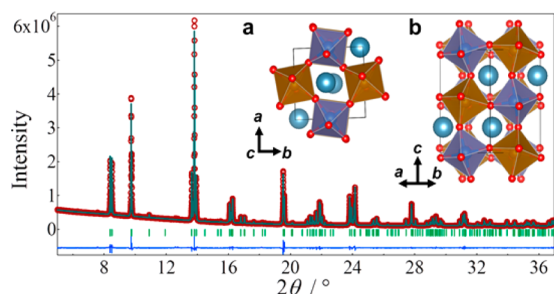


Figure 1. Rietveld refined SXRD profiles of $\text{Ca}_2\text{FeOsO}_6$. Inset (a) is a crystallographic view along the $[001]$ direction and (b) along the $[110]$ direction. The blue and brown octahedral represent FeO_6 and OsO_6 , respectively. The solid spheres represent Ca.

perovskite structure, the degree of the B-site order depends on synthetic conditions ranging from 100% ordered (an ordered rock-salt type) to 0% ordered (a disordered type). Thus, during the refinements, Fe and Os ions were randomly mixed at the same crystallographic site in a 1:1 molar ratio. However, we were unable to reach a reasonable structure solution using these conditions. Alternatively, Fe and Os could be assumed to occupy independent sites. In subsequent refinements, the displacement parameters were temporarily fixed at 0.5 \AA^2 , while the occupancies were fixed. Finally, the analysis indicated that Fe and Os atoms were ordered approximately 90% in the B-site. The small imperfection may be related to the proximity of the ionic radii of Fe^{3+} (0.645 \AA) and Os^{5+} (0.575 \AA).^{28,29} The cell parameters of $\text{Ca}_2\text{FeOsO}_6$ were $a = 5.3931(6) \text{ \AA}$, $b = 5.5084(3) \text{ \AA}$, $c = 7.6790(3) \text{ \AA}$, and $\beta = 89.0211(49)^\circ$. The reliable indexes were $R_p = 1.81\%$ and $R_{wp} = 2.95\%$.³⁰ Details of the analysis are summarized in Tables S1 and S2.

The average bond length of $\text{Os}^{5+}-\text{O}$ was $\sim 1.956 \text{ \AA}$, comparable to that of $\text{Os}^{5+}-\text{O}$ in other perovskite oxides (1.946 \AA for NaOsO_3 ³¹ and 1.960 \AA for $\text{La}_2\text{NaOsO}_6$).³² The bond valence sum, calculated from the bond distances,^{33,34} was 3.00 for Fe and 4.74 for Os, indicating 3+ and 5+ valence states of the atoms, respectively.³³ Bond valence parameters $r_{\text{Fe}-\text{O}} = 1.751$, $r_{\text{Os}-\text{O}} = 1.868$, and $B = 0.37$ were used in the estimation.³⁴

The average bond length of $\text{Fe}^{3+}-\text{O}$ (2.005 \AA) was comparable or slightly larger than that of $\text{Ca}_2\text{FeMoO}_6$ (2.026 \AA) and $\text{Sr}_2\text{FeOsO}_6$ (1.938 \AA); hence, it is reasonable to conclude that Fe^{3+} of $\text{Ca}_2\text{FeOsO}_6$ is in the high-spin state as well as Fe^{3+} of $\text{Ca}_2\text{FeMoO}_6$ ³⁵ and $\text{Sr}_2\text{FeOsO}_6$.³⁶ Note that a high-spin to low-spin state transition via intermediate-spin state of Fe^{3+} in octahedral environment usually occurs at extremely high-pressure conditions far beyond 6 GPa.^{37,38} Comprehensive high-pressure studies of Fe oxides also imply the high-spin state for Fe^{3+} of $\text{Ca}_2\text{FeOsO}_6$.^{37,38}

The crystal structure of $\text{Ca}_2\text{FeOsO}_6$ is shown in Figure 1, based on the experimental solution, showing that Fe and Os ions alternate to occupy the octahedra like a checker board. Bond angles of the interoctahedral Fe–O–Os are 151° and 154° , which are far from 180° , indicating significant buckling of the octahedral connection. Insets a and b show alternate rotations of the octahedra along and perpendicular to the c axis, respectively. The Glazer's notation is $a^-a^+b^+$, where the superscripts indicate that neighbor octahedra rotated in the same (+) and opposite (−) direction.³⁹ The degree of distortion is clearly more enhanced than that of $\text{Sr}_2\text{FeOsO}_6$, where the interoctahedral Fe–O–Os are 180° and 165° .³⁶

Isothermal magnetizations of $\text{Ca}_2\text{FeOsO}_6$ at 2 and 300 K are compared with those of $\text{Sr}_2\text{FeOsO}_6$ in Figure 2. The

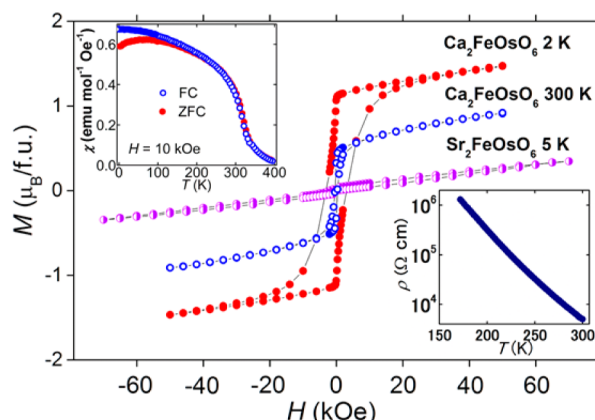


Figure 2. Isothermal magnetizations of $\text{Ca}_2\text{FeOsO}_6$ and $\text{Sr}_2\text{FeOsO}_6$.³⁶ Upper and lower insets show temperature dependent magnetic susceptibility and electrical resistivity of polycrystalline $\text{Ca}_2\text{FeOsO}_6$, respectively.

spontaneous magnetization of $\text{Ca}_2\text{FeOsO}_6$ was $1.2 \mu_B/\text{f.u.}$ at 2 K ($0.5 \mu_B/\text{f.u.}$ at 300 K). This was in large contrast to the near zero magnetization of isoelectronic $\text{Sr}_2\text{FeOsO}_6$.³⁶ The observed magnetization was, however, much lower than the theoretical sum of the spin only moment $8.0 \mu_B/\text{f.u.}$ of Fe^{3+} ($t_{2g}^3 e_g^2$) and Os^{5+} (t_{2g}^3). It was closer to the difference of each spin only moment ($2.0 \mu_B/\text{f.u.}$). Temperature dependent magnetic susceptibility in an applied field of 10 kOe is shown in the upper inset of Figure 2. A remarkable increase was observed approximately 320 K, indicating the occurrence of a long-range magnetic order. Specific heat measurements revealed that it is a bulk transition (Figure S1a). The measurements therefore suggest that an FIM order is established at 320 K with cooling (Figure S2).

A small disagreement between the observed moment ($1.2 \mu_B/\text{f.u.}$) and the theoretical FIM spin-only moment ($2.0 \mu_B/\text{f.u.}$) was partly due to the imperfect order of Fe/Os atoms. The disorder should decrease the magnetic moment as follows: $M = M_{\text{exp}} \times (1 - 2AS)$, where M_{exp} is the expected moment and AS is the degree of disorder between Fe and Os atoms.^{40–42} In $\text{Ca}_2\text{FeOsO}_6$, M_{exp} was $2 \mu_B/\text{f.u.}$ and AS was 0.1 (10% disorder), resulting in $1.6 \mu_B/\text{f.u.}$, approaching the observed moment of $1.2 \mu_B/\text{f.u.}$ In addition, rather stronger hybridization between 5d and oxygen 2p orbitals may reduce the local magnetic moments to some extent, accounting in part for the overestimation of the magnetic moment.

The lower inset of Figure 2 shows the temperature dependent electrical resistivity (ρ) of $\text{Ca}_2\text{FeOsO}_6$. At room temperature, the ρ was approximately $5 \text{ k}\Omega\text{-cm}$ and continued to increase upon cooling beyond the instrumental limit. The seemingly semiconducting temperature dependence was well characterized by the Arrhenius model; the activation energy was estimated to be 1.2 eV within the available temperature range (Figure S3). To further confirm the bulk semiconducting behavior, the electronic structure of $\text{Ca}_2\text{FeOsO}_6$ was investigated by means of first-principle calculations as shown in Figure S4. The result clearly indicated that $\text{Ca}_2\text{FeOsO}_6$ is metallic in nature, contradicting the observation in the ρ vs T measurement. Alternatively, the spin-polarized calculation resulted in a stable solution with a substantial charge gap as

shown in Figure S5. Therefore, an antiparallel arrangement of the Os and Fe moments (the up-spin state of Os and downspin state of Fe are preferably occupied) can be expected as the true ground state of $\text{Ca}_2\text{FeOsO}_6$. The Sommerfeld coefficient was nearly zero in the specific-heat measurements (Figure S1b), supporting the notion of a full charge gap. The theoretical study suggested that the spin-polarization is essential to account for the observed FIM and semiconducting properties.

On the basis of the results above, the most probable magnetic state is proposed in Figure 3a and compared with the

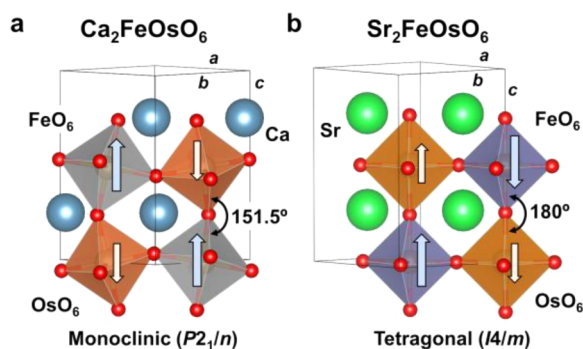


Figure 3. Proposed magnetic ordered state of (a) $\text{Ca}_2\text{FeOsO}_6$ and (b) comparison with the AF1 state of $\text{Sr}_2\text{FeOsO}_6$ (sketched from refs 25 and 36). Note that arrow denotes either up-spin or down-spin.

antiferromagnetic (AF) model proposed for $\text{Sr}_2\text{FeOsO}_6$ (the AF1 phase)^{25,36} seen in Figure 3b. The nearest-neighbor spin interaction is mediated by superexchange interactions via $\text{Fe}^{3+}-\text{O}-\text{Os}^{5+}$ in both compounds.⁴³ Regarding $\text{Sr}_2\text{FeOsO}_6$, it was argued that the superexchange interaction along the c axis was ferromagnetic (FM) because of the nature of the 180° d^3-d^5 bond. An artificial superlattice of $\text{LaFeO}_3/\text{LaCrO}_3$ consisting of a similar 180° d^3-d^5 bond was indeed FM below 375 K.⁴⁴ Meanwhile, the $\text{Fe}-\text{O}-\text{Os}$ bond in the ab plane significantly bends to 165° , leading AF order in the ab plane.⁴³ Very recently, the proposed model of $\text{Sr}_2\text{FeOsO}_6$ was confirmed by neutron diffraction.²⁵ In contrast, $\text{Ca}_2\text{FeOsO}_6$ has $\text{Fe}^{3+}-\text{O}-\text{Os}^{5+}$ bonds at $151-154^\circ$ along and perpendicular to the c axis, respectively. Because the d^3-d^5 superexchange interaction varies from FM to AF in a range of $125-150^\circ$,⁴³ the proposed FIM model does not violate the superexchange model.⁴³

The high- T_c FIM state of $\text{Ca}_2\text{FeOsO}_6$ was adapted from the AF state of $\text{Sr}_2\text{FeOsO}_6$ through a lattice distortion. Thus, the AF to FIM transition driven by lattice distortion was observed for the first time among double perovskite oxides, and is perhaps one of the few examples in all known solid-state oxides. Therefore, $\text{Ca}_2\text{FeOsO}_6$ represents not only a practically useful form of magnetism, but also a scientifically exotic state.

In conclusion, the novel double perovskite oxide $\text{Ca}_2\text{FeOsO}_6$ was synthesized under a pressure of 6 GPa, and it crystallized into a monoclinic structure with a space group of $P2_1/n$. The magnetic and specific-heat measurements characterized the FIM transition at T_c of 320 K. The ρ measurement revealed the electrically insulating state with an activation energy of 1.2 eV, although $\text{Ca}_2\text{FeOsO}_6$ is not a band insulator. The theoretical consideration of the electronic structure draws the highly spin-polarized Fe-3d and Os-5d FIM picture, which is near the HM state [see Figure S5, in which the magnetic moments in atomic spheres are evaluated as $4.02 \mu_B$ for Fe and $-1.46 \mu_B$ for Os without spin-orbit (SO) interaction and $4.02 \mu_B$ for Fe and $-1.34 \mu_B$ for Os with SO interaction]. Although the SO

interaction seems to be insignificant, complex interplays between spins and orbitals likely play an important role in establishing the FIM state of $\text{Ca}_2\text{FeOsO}_6$, unlike what was found in conventional FIM materials such as ferrite and garnet.⁴⁵ This was supported by the drastic impact of the lattice distortion on the magnetism, in addition to the competing spin structures recently discovered in the isoelectronic $\text{Sr}_2\text{FeOsO}_6$.²⁵

$\text{Ca}_2\text{FeOsO}_6$ and $\text{Sr}_2\text{CrOsO}_6$ establish a new class of high- T_c 5d-3d hybrid FIM insulators, which is adjacent to the HM state and hence can be useful for scientific and practical applications in spintronics. The origin of the novel FIM transition with the remarkably higher T_c of 320 K for $\text{Ca}_2\text{FeOsO}_6$ and 725 K for $\text{Sr}_2\text{CrOsO}_6$ remains to be elucidated and further studies including carrier doping to the novel FIM state are in progress as well as neutron diffraction study.

■ ASSOCIATED CONTENT

📄 Supporting Information

Includes synthesis and characterization of the compound. The refined atom coordination, selected bond lengths and angles. Specific heat, inverse magnetic susceptibility, and electrical resistivity analysis. The total density of states of $\text{Ca}_2\text{FeOsO}_6$ based on the GGA, GGA+SOC, and GGA+U calculations, respectively. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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■ REFERENCES

- (1) Imada, M.; Fujimori, A.; Tokura, Y. *Rev. Mod. Phys.* **1998**, *70*, 1039.
- (2) Haskel, D.; Fabbri, G.; Zhernenkov, M.; Kong, P. P.; Jin, C. Q.; Cao, G.; van Veenendaal, M. *Phys. Rev. Lett.* **2012**, *109*, 027204.
- (3) Laguna-Marco, M. A.; Haskel, D.; Souza-Neto, N.; Lang, J. C.; Krishnamurthy, V. V.; Chikara, S.; Cao, G.; van Veenendaal, M. *Phys. Rev. Lett.* **2010**, *105*, 216407.
- (4) Kobayashi, K. L.; Kimura, T.; Sawada, H.; Terakura, K.; Tokura, Y. *Nature* **1998**, *395*, 677.
- (5) Sarma, D. D.; Mahadevan, P.; Saha-Dasgupta, T.; Ray, S.; Kumar, A. *Phys. Rev. Lett.* **2000**, *85*, 2549.

- (6) Tomioka, Y.; Okuda, T.; Okimoto, Y.; Kumai, R.; Kobayashi, K. I.; Tokura, Y. *Phys. Rev. B* **2000**, *61*, 422.
- (7) Jeng, H. T.; Guo, G. Y. *Phys. Rev. B* **2003**, *67*, 094438.
- (8) Kobayashi, K. I.; Kimura, T.; Tomioka, Y.; Sawada, H.; Terakura, K.; Tokura, Y. *Phys. Rev. B* **1999**, *59*, 11159.
- (9) Solov'yev, I. V. *Phys. Rev. B* **2002**, *65*, 144446.
- (10) Kanamori, J.; Terakura, K. *J. Phys. Soc. Jpn.* **2001**, *70*, 1433.
- (11) Erten, O.; Meetei, O. N.; Mukherjee, A.; Randeria, M.; Trivedi, N.; Woodward, P. *Phys. Rev. Lett.* **2011**, *107*, 257201.
- (12) Chmaissem, O.; Dabrowski, B.; Kolesnik, S.; Short, S.; Jorgensen, J. D. *Phys. Rev. B* **2005**, *71*, 174421.
- (13) Popov, G.; Greenblatt, M.; Croft, M. *Phys. Rev. B* **2003**, *67*, 024406.
- (14) Arulraj, A.; Ramesha, K.; Gopalakrishnan, J.; Rao, C. N. R. *J. Solid State Chem.* **2000**, *155*, 233.
- (15) Wu, H. *Phys. Rev. B* **2001**, *64*, 125126.
- (16) Serrate, D.; De Teresa, J. M.; Ibarra, M. R. *J. Phys.: Condens. Matter* **2007**, *19*, 023201.
- (17) Vaitheeswaran, G.; Kanchana, V.; Delin, A. *Appl. Phys. Lett.* **2005**, *86*, 032513.
- (18) Kato, H.; Okuda, T.; Okimoto, Y.; Tomioka, Y.; Takenoya, Y.; Ohkubo, A.; Kawasaki, M.; Tokura, Y. *Appl. Phys. Lett.* **2002**, *81*, 328.
- (19) Philipp, J. B.; Majewski, P.; Alff, L.; Erb, A.; Gross, R.; Graf, T.; Brandt, M. S.; Simon, J.; Walther, T.; Mader, W.; Topwal, D.; Sarma, D. D. *Phys. Rev. B* **2003**, *68*, 144431.
- (20) Philipp, J. B.; Reisinger, D.; Schonecke, M.; Marx, A.; Erb, A.; Alff, L.; Gross, R.; Klein, J. *Appl. Phys. Lett.* **2001**, *79*, 3654.
- (21) De Teresa, J. M.; Serrate, D.; Blasco, J.; Ibarra, M. R.; Morellon, L. *Phys. Rev. B* **2004**, *69*, 144401.
- (22) Kato, H.; Okuda, T.; Okimoto, Y.; Tomioka, Y.; Oikawa, K.; Kamiyama, T.; Tokura, Y. *Phys. Rev. B* **2002**, *65*, 144404.
- (23) Prellier, W.; Smolyaninova, V.; Biswas, A.; Galley, C.; Greene, R. L.; Ramesha, K.; Gopalakrishnan, J. *J. Phys.: Condens. Matter* **2000**, *12*, 965.
- (24) Krockenberger, Y.; Mogare, K.; Reehuis, M.; Tovar, M.; Jansen, M.; Vaitheeswaran, G.; Kanchana, V.; Bultmark, F.; Delin, A.; Wilhelm, F.; Rogalev, A.; Winkler, A.; Alff, L. *Phys. Rev. B* **2007**, *75*, 020404.
- (25) Meetei, O. N.; Erten, O.; Randeria, M.; Trivedi, N.; Woodward, P. *Phys. Rev. Lett.* **2013**, *110*, 087203.
- (26) Abakumov, A. M.; Shpanchenko, R. V.; Antipov, E. V.; Lebedev, O. I.; VanTendeloo, G. *J. Solid State Chem.* **1997**, *131*, 305.
- (27) Battle, P. D.; Jones, C. W.; Studer, F. J. *Solid State Chem.* **1991**, *90*, 302.
- (28) Anderson, M. T.; Greenwood, K. B.; Taylor, G. A.; Poeppelmeier, K. R. *Prog. Solid State Chem.* **1993**, *22*, 197.
- (29) Shannon, R. D. *Acta Crystallogr., A* **1976**, *32*, 751.
- (30) Momma, K.; Izumi, F. *J. Appl. Crystallogr.* **2008**, *41*, 653.
- (31) Shi, Y. G.; Guo, Y. F.; Yu, S.; Arai, M.; Belik, A. A.; Sato, A.; Yamaura, K.; Takayama-Muromachi, E.; Tian, H. F.; Yang, H. X.; Li, J. Q.; Varga, T.; Mitchell, J. F.; Okamoto, S. *Phys. Rev. B* **2009**, *80*, 161104.
- (32) Gemmill, W. R.; Smith, M. D.; Prozorov, R.; Loye, H. C. Z. *Inorg. Chem.* **2005**, *44*, 2639.
- (33) Brese, N. E.; Okeeffe, M. *Acta Crystallogr., B* **1991**, *47*, 192.
- (34) Brown, I. D.; Altermatt, D. *Acta Crystallogr., B* **1985**, *41*, 244.
- (35) Alonso, J. A.; Casais, M. T.; Martinez-Lopez, M. J.; Martinez, J. L.; Velasco, P.; Munoz, A.; Fernandez-Diaz, M. T. *Chem. Mater.* **2000**, *12*, 161.
- (36) Paul, A. K.; Jansen, M.; Yan, B. H.; Felser, C.; Reehuis, M.; Abdala, P. M. *Inorg. Chem.* **2013**, *52*, 6713.
- (37) Ju, S.; Cai, T. Y.; Lu, H. S.; Gong, C. D. *J. Am. Chem. Soc.* **2012**, *134*, 13780.
- (38) Ding, Y.; Haskel, D.; Ovchinnikov, S. G.; Tseng, Y.-C.; Orlov, Y. S.; Lang, J. C.; Mao, H.-k. *Phys. Rev. Lett.* **2008**, *100*, 045508.
- (39) Glazer, A. M. *Acta Crystallogr., B* **1972**, *28*, 3384.
- (40) Serrate, D.; De Teresa, J. M.; Ibarra, M. R. *J. Phys.: Condens. Matter* **2007**, *19*, 023201.
- (41) Alonso, J. L.; Fernandez, L. A.; Guinea, F.; Lesmes, F.; Martin-Mayor, V. *Phys. Rev. B* **2003**, *67*, 214423.
- (42) Michalik, J. M.; De Teresa, J. M.; Ritter, C.; Blasco, J.; Serrate, D.; Ibarra, M. R.; Kapusta, C.; Freudenberger, J.; Kozlova, N. *Europhys. Lett.* **2007**, *78*, 17006.
- (43) Goodenough, J. B. *Magnetism and the Chemical Bond*; Interscience: New York, 1963.
- (44) Ueda, K.; Tabata, H.; Kawai, T. *Science* **1998**, *280*, 1064.
- (45) Spaldin, N. A. *Magnetic Materials: Fundamentals and Applications*, 2nd ed.; Cambridge: Cambridge University Press, 2010; pp 113–129.